High-temperature calorimetric studies on the thermal properties of $Rb_2U(SO_4)_3(s)$

K. Jayanthi, V.S. Iyer, K.D. Singh Mudher and V. Venugopal*

Fuel Chemistry Division, Bhabha Atomic Research Centre, Trombay, Bombay-400 085 (India)

(Received 14 January 1993; accepted 7 April 1993)

Abstract

Measurements of the enthalpy increment, $(H_T^{\ominus} - H_{29815}^{\ominus})$ for Rb₂U(SO₄)₃ were carried out in the range 370-800 K using a high-temperature Calvet calorimeter. Solid state transitions were observed at 625 ± 3 K and 763 ± 2 K, with respective enthalpies of transitions of 5.998 and 13.585 kJ mol⁻¹. The $H_T^{\ominus} - H_{29815}^{\ominus}$ values were plotted against temperature and can be given in J mol⁻¹ by $H_T^{\ominus} - H_{29815}^{\ominus} \pm 694/(J \text{ mol}^{-1}) = -1.117 \times 10^5 + 368.9T/\text{K} + 0.0199T^2/\text{K}^2$ (from 370 to 628 K); $H_T^{\ominus} - H_{29815}^{\ominus} \pm 405/(J \text{ mol}^{-1}) =$ $3.279 \times 10^3 + 13.4T/\text{K} + 0.3063T^2/\text{K}_2$ (from 630 to 760 K); and $H_T^{\ominus} - H_{29815}^{\ominus} \pm 433/(J \text{ mol}^{-1}) = -6.671 \times 10^5 + 1140T/\text{K}$ (from 765 to 803 K).

The thermodynamic properties of $Rb_2U(SO_4)_3(s)$ were derived using the estimated value of $S_{298,15}^{\leftrightarrow}$.

INTRODUCTION

Quality control of nuclear fuel is of prime importance in the nuclear industry for the safe and sustained operation of nuclear reactors. This demands strict control on chemical specifications. In order to check the accuracy of the analytical methods it is essential to employ standard reference materials which have constant stoichiometry, and good stability and solubility. Rubidium uranium trisulphate ($Rb_2U[SO_4]_3$) was suggested as a chemical standard for the determination of uranium [1]. U_3O_8 [2] and high-purity uranium metal [3] are available from NIST as a chemical standard reference material for uranium. Some physicochemical properties of these compounds have been studied by Singh Mudher et al. [4]. Studies on the thermal properties have not been reported so far. The thermal properties of $Rb_2U(SO_4)_3$ are important to characterize the material for its stability at high temperatures. Hence, enthalpy increment measurements

^{*} Corresponding author.

on Rb₂U(SO₄)₃ were carried out using a high-temperature Calvet calorimeter and other thermal properties were derived from the $H_T^{\ominus} - H_{298\,15}^{\ominus}$ values.

EXPERIMENTAL

Materials

TABLE 1

The double sulphate of rubidium and uranium, $(Rb_2U[SO_4]_3)$, was obtained by crystallization from aqueous solution. Uranyl sulphate solution was obtained by the dissolution of nuclear-grade UO₃ in dilute H₂SO₄ and the UO_2^{2+} state was reduced to U^{4+} electrolytically. A solution of Rb_2CO_3 (99% pure from Aldrich Chemical) in $0.5 \text{ M H}_2\text{SO}_4$ was then added to the uranium solution and the resultant solution was concentrated gradually to obtain green crystals of $Rb_2U(SO_4)_3$ [1]. Chemical reagents such as H_2SO_4 , HNO₃ and NH₄OH used for the preparation of $Rb_2U(SO_4)_3$ were of analytical reagent grade. The impurity content in the sample was analysed by atomic emission spectrometry. The total impurity content in the sample was less than 300 ppm as shown in Table 1. The uranium and sulphate contents were analysed chemically. The analysis of uranium was carried out using the Davies and Gray method [5]. The sulphate content was analysed gravimetrically as BaSO₄ after separating uranium from the solution as hydroxide [1]. The concentrations of uranium and sulphate determined in the $Rb_2U(SO_4)_3$ samples were 34.12 ± 0.12 wt.% and 41.28 ± 0.14 wt.%, respectively. The expected theoretical concentrations were 34.15 wt.% and 41.32 wt.%. The X-ray powder diffraction pattern taken on a Siemens diffractometer using Ni-filtered Cu K α radiation, agreed with that reported earlier. $Rb_2U(SO_4)_3$ samples were made in the form of pellets of 3 mm in diameter, and 2 mm in thickness at a pressure of 50 MPa. The pellets were

Element	ppm	Element	ppm	
Al	70	В	<0.1	
Be	< 0.3	Ca	<10	
Cd	<0.1	Cr	12	
Cu	<5	Fe	<20	
Mg	<5	Мо	40	
Na	20	Ni	<5	
Si	<60	Sn	<2	
v	27	Zn	<10	
Со	<5	Mn	<2	
Pb	<5	W	<40	

Results of the impurity analysis of Rb₂U(SO₄)₃

dried at 400 K for 2-3 h and stored in a desiccator before use for high-temperature enthalpy increment measurement.

Thermal analysis

Simultaneous thermogravimetric (TG) and differential thermal analysis (DTA) were carried out on $Rb_2U(SO_4)_3$ samples using an ULVAC Sinku Riko (model TGD-7000) thermal analyser. The calibration of the DTA apparatus was performed using high-purity $K_2CrO_4(s)$. The temperature calibration was carried out by determining the melting point of high-purity tin. Platinum cups were used as container material for the sample and reference. The sample size used for our experiment was 100 mg. Experiments were carried out in both the heating and cooling modes. A heating rate of 10 K min⁻¹ and a cooling rate of 5 K min⁻¹ were used for the studies. Preheated high-purity alumina was used as reference material. The temperatures were scanned from 300 to 973 K.

A Netzsch differential scanning calorimeter was also employed for confirming the transition temperatures of the sample. A heating rate of 15 K min^{-1} and sample size of 50 mg were used in the experiments.

The Calvet calorimetric system and experimental procedure

The enthalpy increment measurements were carried out using a Setaram high-temperature Calvet microcalorimeter. Details of the experimental procedure and calibration of the calorimeter were reported in earlier publications [6, 7]. The Calvet calorimeter has identical twin compartments surrounded by thermopiles kept inside a massive fritted alumina calorimetric block whose temperature is monitored and controlled. The calorimetric block is heated by a Kanthal wound furnace. Two closed-end long alumina tubes are inserted into the two compartments to act as sample and reference holders, respectively. The closed-end alumina tube helps in maintaining any type of gas environment over the sample. The compartments exchange heat with the block. Thermopiles surrounding the two compartments are connected in opposition to minimize thermal disturbance. The differential signal from the thermopile is proportional to the heat evolved/absorbed in any one compartment. This exchange of heat gives an electrical signal from the thermopile which is amplified and fed into a recorder or a computer for data acquisition and processing. The twin alumina compartments were evacuated to 100 Pa and flushed with high-purity argon before the start of the experiment. The enthalpy increment measurements were carried out by dropping the samples from a thermostated enclosure kept at 298.15 K into the closed-end alumina tube. The measurements were made at different isothermal temperatures from 370 to 800 K at intervals of 10-15 K. At each isothermal temperature, two or three samples were dropped one after another to obtain precise values. The accuracy of the measurements was checked by dropping NIST alumina samples (synthetic sapphire SRM-720) at the beginning and end of each experiment. From the calibration runs, it was found that the precision of the measurements was $\pm 0.5\%$ and accuracy was better than $\pm 0.5\%$.

RESULTS

Thermal analysis

Figure 1 shows the DTA annd TG curves for $Rb_2U(SO_4)_3$. Two endotherms are indicated, one at 628 K and another at 780 K, before the decomposition starts at 893 K. TG curve shows a weight loss in the range 893–953 K. The weight loss can be attributed to the loss of SO₂ according to the reaction given below. The products $Rb_2UO_2(SO_4)_2$ and SO₂ were not confirmed by any chemical analysis

 $Rb_2U(SO_4)_3 \rightarrow Rb_2UO_2(SO_4)_2 + SO_2$

The expected weight loss according to the above reaction is 9.18 wt.% which is in very good agreement with the observed value of 9.17 wt.% in the TG curve.



Fig. 1. DTA and TG traces showing the transitions and decomposition of $Rb_2U[SO_4]_2$. Sample weight 517 mg.



Fig. 2. DSC curves showing the transitions of $Rb_2U(SO_4)_3$.

Figure 2 shows the differential scanning calorimetric results. The first endothermic peak is at 616 K and the second at 773 K. The third peak in the DSC curve at 898 K corresponds to the decomposition of $Rb_2U(SO_4)_3$.

Calorimetric results

The enthalpy increment values of $Rb_2U(SO_4)_3$ at various temperatures are given in Table 2. The enthalpy increments were least-squares fitted with temperature into three separate linear segments. The equations obtained were

$$H_T^{\oplus} - H_{298\,15}^{\oplus} \pm \frac{694}{(\text{J mol}^{-1})} = -1.117 \times 10^5 + \frac{368.97}{\text{K}} + 0.01997^2/\text{K}^2$$

(from 370 to 628 K)

$$H_T^{\ominus} - H_{298\,15}^{\ominus} \pm 405/(\mathrm{J\,mol^{-1}}) = 3.279 \times 10^3 + 13.4T/\mathrm{K} + 0.3063T^2/\mathrm{K}^2$$

(from 630 to 760 K)

$$H_T^{\ominus} - H_{298,15}^{\ominus} \pm 433/(\text{J mol}^{-1}) = -6.671 \times 10^5 + 1140T/\text{K}$$

(from 765 to 800 K)

The dependence of the enthalpy increment values on temperature is compared with the fit values in Fig. 3. It can be seen from the fit that there are two solid state transitions: one at 625 ± 3 K and the other at 763 ± 2 K. The corresponding enthalpies of transition were calculated as 5.998 and 13.585 kJ mol⁻¹, respectively.

T in	$H_T^{\leftrightarrow} - H_{298,15}^{\ominus}$	T in	$H_T^{\ominus} - H_{29815}^{\ominus}$	
Κ	in $J \mod^{-1}$	К	in $\mathbf{J} \mathbf{mol}^{-1}$	
298.15	0.0	First transiti	on	
370.2	26717 ± 150	630.0	133291 ± 450	
388.4	33681 ± 180	639.8	137268 ± 340	
418.0	45850 ± 276	658.2	144756 ± 876	
437.6	52072 ± 325	668.5	148919 ± 610	
458.1	60091 ± 432	678.6	152985 ± 241	
498.1	77112 ± 310	693.2	160068 ± 720	
518.6	85018 ± 543	708.3	167147 ± 870	
539.0	93256 ± 498	752.5	185656 ± 900	
559.7	100608 ± 870	760.0	190381 ± 1020	
580.2	109278 ± 765			
593.6	112436 ± 981	Second trans	sition	
604.9	117789 ± 1076	763.7	203966 ± 1015	
617.1	123162 ± 670	773.0	213847 ± 1190	
623.2	125479 ± 698	788.0	231062 ± 786	
625.4	127293 ± 1439	803.0	248643 ± 943	

TABLE 2

Enthalpy increments of Rb₂U(SO₄)₃ at various temperatures



Fig. 3. Dependence of enthalpy increment on temperature for $Rb_2U(SO_4)_3(s)$ samples.

T in K	$H_T^{\ominus} - H_{29815}^{\ominus}$ in J mol ⁻¹	C_{ρ}^{\ominus} in JK ⁻¹ mol ⁻¹	S_T^{\leftrightarrow} in $\mathbf{J}\mathbf{K}^{-1}\mathbf{mol}^{-1}$	$-(G_T^{\leftrightarrow} - 298.15)/T$ in JK ⁻¹ mol ⁻¹
325	9598	384.9	384.9	354.6
350	19232	385.7	412.7	367.8
375	28886	386.5	439.4	362.3
400	38559	387.3	464.3	367.9
425	48251	388.1	487.8	374.3
450	57962	388.8	510.0	381.2
475	67692	389.6	531.1	388.6
500	77442	390.4	551.1	396.2
525	87211	391.1	570.1	404.0
550	96999	391.9	588.4	412.0
575	106806	392.7	605.8	420.1
600	116633	393.4	622.5	428.1
625	126478	394.2	638.6	436.2
Transitio	n			
650	141401	411.6	577.0	359.5
675	151882	426.9	592.8	367.8
700	162746	442.2	608.6	376.1
725	173993	457.5	624.4	384.4
750	185623	472.9	640.2	392.7

Thermal properties of $Rb_2U(SO_4)_3(s)$

TABLE 3

The first differential of the enthalpy equations with respect to temperature yields the molar heat capacity. The S_{29815}^{\ominus} value of $Rb_2U(SO_4)_3$ is assumed to be the summation of S_{29815}^{\ominus} values of $Rb_2SO_4(s)$ and $U(SO_4)_2$ reported in the literature [8]. From this value, the S_T^{\ominus} of $Rb_2U(SO_4)_3$ is calculated using the equation

$$S_T^{\oplus} = S_{298\,15}^{\oplus} + \int_{298\,15}^T (C_p^{\oplus}/T) \,\mathrm{d}T$$

The Gibbs free energy function $-(G_T^{\ominus} - H_{298.15}^{\ominus})/T$ was derived from the equation

$$-(G_T^{\ominus} - H_{298\,15}^{\ominus})/T = S_T^{\ominus} - [(H_T^{\ominus} - H_{298\,15}^{\ominus})/T]$$

The calculated values of the thermal properties are given in Table 3.

DISCUSSION

The transition temperatures obtained by calorimetry are in reasonable agreement with the DSC and DTA results. A difference of 10 K in the second transition temperature between DSC and calorimetry may be attributed to the dynamic nature of DSC in which to some extent, the peak temperature is a function of heating rate. The decomposition temperatures obtained in DSC and DTA are in very good agreement with each other. The solid state transitions were found to be reversible. The X-ray diffraction pattern before and after the calorimetric experiments below 800 K remained the same, indicating no basic structural changes during the calorimetric study.

ACKNOWLEDGEMENTS

The authors thank Dr. D.D. Sood, Associate Director and Head, Fuel Chemistry Division, for his keen interest in this work, and Dr. N.C. Jayadevan for his help in the XRD studies.

REFERENCES

- 1 K.D. Singh Mudher, R.R. Khandekar, K. Krishnan, N.C. Jayadevan, and D.D. Sood, B.A.R.C. Report-1476, 1989.
- 2 G.S. Petit and C.A. Kienberger, Anal. Chim. Acta, 25 (1961) 579.
- 3 K.F. Lauer and Y. Le Duigou, Proc. Symp. Anal. Methods in Nucl. Fuel Cycle, Vienna, 29 November-3 December 1971, IAEA, 1972, p. 145.
- 4 K.D. Singh Mudher, K. Krishnan, D.M. Chackraburtty and N.C. Jayadevan, J. Less Common Met., 143 (1988) 173.
- 5 W. Davies and W. Gray, Talanta, 11 (1964) 1203.
- 6 V.S. Iyer, Renu Agarwal, K.N. Roy, S. Venkateswaran, V. Venugopal and D.D. Sood, J. Chem. Thermodyn., 439 (1990) 22.
- 7 R. Prasad, Renu Agarwal, K.N. Roy, V.S. Iyer, V. Venugopal and D. D. Sood, J. Nucl. Mater., 167 (1989) 261.
- 8 O. Kubaschewski and C.B. Alcock, Metallurgical Thermochemistry, Pergamon Press, Oxford, 5th edn., 1979.